

# HIGH QUALITY (100) AND (001) ORIENTED SUBSTRATES PREPARED FROM CZOCHRALSKI GROWN SrLaGaO<sub>4</sub> AND SrLaAlO<sub>4</sub> SINGLE CRYSTALS

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The growth of SrLaGaO<sub>4</sub> and SrLaAlO<sub>4</sub> crystals on  $\langle 100 \rangle$  and  $\langle 001 \rangle$  oriented seeds was investigated. Various defects, which appeared in crystals grown on these two orientations, were observed in polarized light and by X-ray diffraction topography. It was found that to obtain a substrate of the best quality, the crystal should be cut along the growth directions. Therefore, crystals pulled along  $\langle 100 \rangle$  direction are utilized for preparation of (001) substrates, whereas (100) substrates are better to cut from crystals grown on  $\langle 001 \rangle$  seed. The quality of the prepared substrates was determined by high resolution X-ray diffraction study in terms of rocking curve and mean mosaic angle.

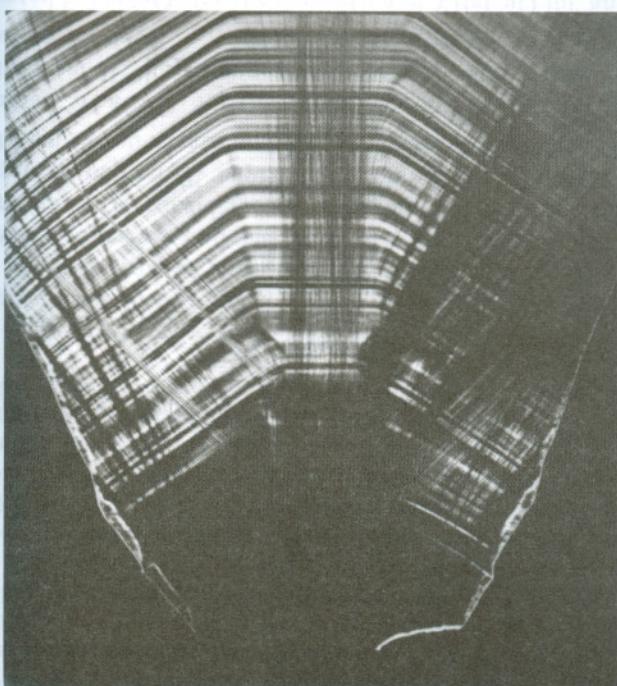
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## 1. Introduction

Single crystals with tetragonal structure of K<sub>2</sub>NiF<sub>4</sub>-type (*I4/mmm*) were proposed for the first time as substrates for HTSC epitaxy [1] because of their suited dielectric properties and the lack of phase transitions. SrLaAlO<sub>4</sub> and SrLaGaO<sub>4</sub> are currently considered [2-6] as some of the best substrate materials for the epitaxy of high-*T<sub>c</sub>* superconductors. Taking into account the small mismatch between the substrate and HTSC layers, the application areas include:

- (001) oriented SrLaGaO<sub>4</sub> for the deposition of YBaCuO [2, 3], and 1212 and 1223 phases of Hg based superconductors [4] with *c*-axis perpendicular to the substrate surface,
- (100) oriented SrLaGaO<sub>4</sub> for the deposition of YBaCuO [5], and 1212 mercury films with *c*-axis parallel to the substrate surface,
- (001) oriented SrLaAlO<sub>4</sub> [6] for the deposition of La<sub>2-x</sub>Sr<sub>x</sub>CuO<sub>4</sub>.

Fig. 1. X-ray diffraction topography. Slice cut parallel to the (001) plane from the initial part of SrLiAlO<sub>4</sub> crystal grown on (100) oriented seed.



Single crystal obtained on a (100) oriented seed has almost elliptical cross section. At its circumference, parallel to the longer axis of this ellipse two (001) facets occur. On the remaining part of the crystal axis two (001) facets are much less developed than the (001) planes. In the central part of the crystal grown on a (100) oriented seed two types of (101) planes appear. They are much less developed than the (001) planes. In the central part of the crystal plane (001) planes corresponds to the crystal-melt interface. The boundary between these planes is the crystal-grown boundary. In the central part of the crystal plane (101) planes corresponds to the crystal-grown boundary. The boundary between these planes is the crystal-melt interface. Various defects, which appear in crystals grown with the crystal-grown boundary, are not observed in crystals grown with the crystal-melt interface.

## 2. Experiment and results

orientation, were observed in polarized light as well as by X-ray diffraction topography. For the sample cut from the initial part of the  $\text{SrLaAlO}_4$  crystal parallel to the (001) plane, the growth striations, indicating the shape of crystal-melt interface, are shown in Fig. 1. In the core region many dislocations, propagating from the seed along the crystal growth direction, are visible. Crystal quality in the core region is closely related to the seed quality and the procedure used to the start of the crystal growth. Dislocations and defects of the crystal structure in the seed used for crystal growth are very well visible in core region both, in samples cut parallel and perpendicular to the growth direction. X-ray diffraction topography of slices cut from the crystal perpendicularly to the growth direction shows the core region in the central part of the crystal, very often with well visible lines corresponding to stacking faults. Internal stresses resulting from the difference in the chemical composition of core and faceted regions sometimes cause small cracks parallel to the cleavage plane.

The shape of the cross-section of a single crystal obtained on (001) oriented seed is a square with cut corners. The larger planes of the crystal surface correspond

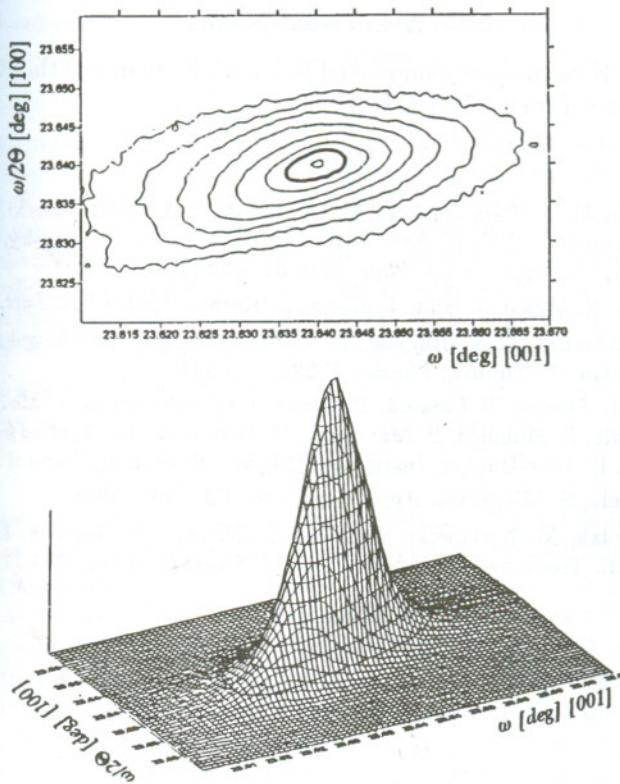


Fig. 2. Reciprocal lattice mapping of  $\text{SrLaGaO}_4$  (100) oriented substrate, (200) reflection,  $\text{Cu } K_\alpha$  radiation.

to the (100) planes and the smaller ones to (110). In the crystal grown on (001) oriented seed one (001) plane and four (103) type planes appear on the crystal-melt interface. In this crystal the core region is formed by the central part grown on the (001) plane whereas the outside faceted region consists of four (103)-type planes.

The quality of the prepared substrates was determined by high resolution X-ray diffraction study in terms of rocking curve and mean mosaic angle. Typical results, obtained on the  $\text{SrLaGaO}_4$  crystal with the substrate orientation (100) are presented in Fig. 2. The theoretical value of the full width at half maximum of the rocking curve, obtained for the (200) reflex is 12.0 arcsec and for (006) reflex it is 9.3 arcsec, the experimental values measured on (100) substrates are in the range of 13 to 24 arcsec and on (001) substrates are 13 to 22 arcsec. The experimental values of mean mosaic angle for (100) and (001) substrates are in the range of 25 arcsec to 30 arcsec and 20.5 arcsec to 23 arcsec, respectively.

For single crystals with small core dimensions it was found that substrates of the best quality can be obtained when the crystal is cut along the growth direction from the faceted region. Therefore, crystals pulled along the  $\langle 100 \rangle$  direction are used for preparation of (001) substrates, whereas the (100) substrates should be cut from crystals grown on  $\langle 001 \rangle$  seed.

### Acknowledgments

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