

Fast GaInAsSb p-i-n photodiode for the spectral interval 2-3 μm

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Loss of less than 0.01 dB/km is expected¹ in optical-fiber communication lines based on fibers operating in the spectral range 2-4 μm . This loss is an order of magnitude lower than quartz fibers in the range 1.3-1.6 μm . The loss which has been achieved in the fabrication of fibers² and a report of the first commercial structures of a fluoride fiber³ are stimulating interest in the development of the basic elements (sources and photodetectors) of a third generation for infrared optical-fiber communications in this spectral range.

The development of efficient injection lasers on GaInAsSb solid solutions for the wavelength interval 2-2.4 μm has been reported in recent years (4-6). Among these lasers are some which can operate continuously at room temperature.⁷ There have also been some studies of photodetectors based on multicomponent III-V solid solutions for this wavelength interval.^{8,9}

We have previously reported⁵ the fabrication and study of the first uncooled photodiodes with a quantum efficiency ~ 0.6 in the range 1.4-2.4 μm and a response time $\tau \approx 0.5$ ns as well as avalanche photodiodes¹⁰ based on GaInAsSb/GaAlAsSb solid solutions.

In the present letter we are reporting the results of the fabrication and study of ultrafast p-i-n photodiodes based on GaInAsSb.

The structures were grown by liquid-phase epitaxy on a p-type GaSb substrate with a carrier concentration $p = 2 \cdot 10^{17} \text{ cm}^{-3}$. A layer of the GaInAsSb solid solution 2 μm thick, with a band gap $E_g = 0.53$ eV, not deliberately doped, was covered by a wide-gap layer of n^+ -GaAlAsSb with concentration $n = 10^{18} \text{ cm}^{-3}$ ($E_g = 1.2$ eV) and a thickness of 0.3 μm .

The carrier concentration in the GaInAsSb narrow-gap active layer was $p^0 = 4.8 \cdot 10^{14} \text{ cm}^{-3}$, according to measurements of the voltage dependence of the capacitance. This concentration, we might note, is an order of magnitude lower than that reported in Refs. 8 and 10. The intrinsic concentration in the GaInAsSb solid solution with $E_g = 0.53$ eV is $n_i = 4 \cdot 10^{13} \text{ cm}^{-3}$.

The photodiodes were fabricated by photolithography in the form of mesa diodes with a mesa diameter of 90 μm . The contacts were formed by depositing a Au:Zn alloy on the p-GaSb substrate and a Au:Te alloy on the wide-gap window, with subsequent deposition of a layer of Ag. The structure of the p-i-n diode is shown schematically in Fig. 1a.

We studied the current-voltage and capacitance-voltage characteristics of the diodes, the sensitivity spectrum, and the speed of the p-i-n

diodes by the method of pulsed scanning electron microscopy.

The plot of the capacitance versus the voltage was characteristic of p-i-n diodes. The minimum capacitance in the depletion regime was $C = 0.4$ pF at a bias voltage $V = 2$ V. The width of the space-charge region was 2 μm , which corresponded to the thickness of the GaInAsSb narrow-gap layer. This low capacitance was achieved by virtue of the low carrier concentration in this layer. The dark current at a voltage of 0.5 V was 8-10 μA and was associated with the generation-recombination current in the space-charge layer.

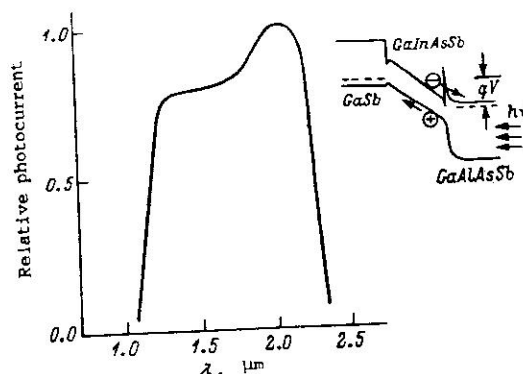


FIG. 1. Photosensitivity spectrum of a GaInAsSb/GaAlAsSb p-i-n photodiode at $T = 300$ K. The inset is a schematic band diagram of the p-i-n structure with a reverse bias.

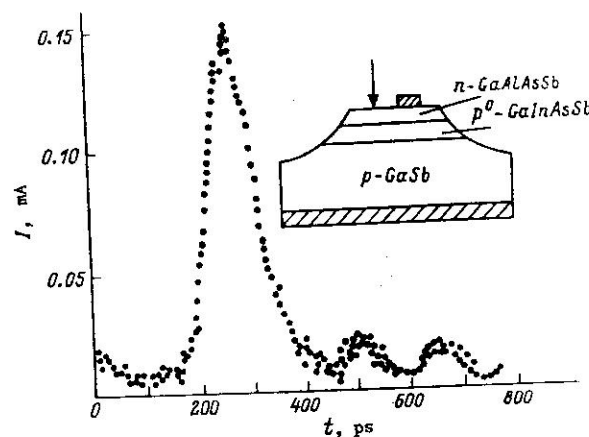


FIG. 2. Kinetics of the current induced by the electron probe with pulsed excitation of p-i-n structure. The inset shows the GaInAsSb/GaAlAsSb structure in the p-i-n photodiode.

Spectra of the photoresponse were recorded on an SPM-2 monochromator with a LiF prism.

Figure 1 shows the photosensitivity spectrum of a GaInAsSb/GaAlAsSb p-i-n diode at room temperature. The magnitude of the photocurrent is essentially independent of the applied voltage. The monochromatic sensitivity at the wavelength $\lambda = 2-2.2 \mu\text{m}$ was $\sim 1.0 \text{ A/W}$. The size of the sensitive area of the p-i-n diode was $6 \cdot 10^{-5} \text{ cm}^2$. The apparent reason for the decay of the photoresponse in the short-wavelength part of the spectrum is the additional absorption in the heavily doped N^+ -GaAlAsSb covering layer, 3 μm thick.

We studied the speed of the p-i-n diodes by the method of pulsed scanning microscopy. The experimental apparatus included an REM-100U scanning electron microscope with a picosecond electron-beam modulator, an SI-91/3 sampling oscilloscope controlled by an MERA-60 computer, a microwave sample holder, and an in-vacuum system for bringing out the signals for recording.¹¹ The minimum pulse length was 25-35 ps, depending on the beam current.

The test diodes were mounted in a gap in a 50- Ω stripline, from which the output signals was fed through a coaxial-stripline adapter section to the input of the sampling oscilloscope. These measurements were carried out at an accelerating voltage of 45 kV. Calculations¹² using the electron-hole-pair generation function showed that no more than 20% of the current carriers were excited in the quasineutral regions of the diodes adjacent to the space-charge regions. The peak value of the beam current in the pulse was $5 \cdot 10^{-8} \text{ A}$ which corresponds to an irradiation power $\sim 1 \text{ mW}$ in a local region about 2 μm in size on the diode. The pulse repetition rate was in the range 30-100 KHz.

Figure 2 shows results of a kinetic study of the photoresponse in local regions of an InGaAsSb/GaAlAsSb diode structure during the application of a reverse bias voltage $V = 4 \text{ V}$. The pulse rise time ($\tau_r = 50 \text{ ps}$) is determined by a convolution of three time scales: the length of the exciting pulse, the RC component (the capacitance in this diode construction was $C = 0.4 \text{ pF}$), and the rise time of the transient characteristic of the oscilloscope ($\sim 30 \text{ ps}$). The decay time is limited by — in addition to the factors just cited — the carrier transit time in the active region of the p-i-n diode. Taking the approach of Ref. 13, we can estimate the transit time to be $t_{tr} = 45-50 \text{ ps}$, which corresponds to a carrier drift velocity $v_{dr} = (5-6) \cdot 10^6 \text{ cm/s}$ in fields $E \sim 2 \cdot 10^{14} \text{ V/cm}$ (the carriers are presumably holes). The drift velocity has not been measured previously in GaInAsSb solid solutions. The photosensitivity and the speed were uniform within 5% over the area of the structures.

In these p-i-n photodiodes we did not observe any long-term showing of the transient response in

the region of the decay of the photoresponse pulse, which is an important problem in the fabrication of fast InGaAs/InP p-i-n and avalanche photodiodes. The reason lies in the particular features of the type-II heterostructure in the p-GaSb-p⁺-GaInAsSb-n-GaAlAsSb structure which was used. It can be seen from the schematic band diagram in the inset in Fig. 1 that during the application of a blocking voltage the conditions for a pile-up of holes at the heterojunction are not satisfied in a structure of this type.

This study has resulted in the fabrication of the first p-i-n photodiode with a very high speed ($\tau = 50 \text{ ps}$) on the basis of GaInAsSb narrow-gap solid solutions for the spectral region 2-2.4 μm . The speed is comparable to the better results which have been obtained to date for the well-developed GaInAs/InP p-i-n diodes at a wavelength of 1.3 μm (Ref. 15, for example). Such p-i-n photodiodes hold promise for use in the reception modules of infrared optical-fiber communications at high data transmission rates, on the order of several gigabits/s. They may also find use for recording fast processes in scientific research.

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